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Improvement of Cu(In,Ga)(S,Se)₂ thin film Solar Cells with the help of Gallium and Sulfur Gradients

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Das Ziel der Arbeit besteht darin, optimierte Absorber für das Dünnschicht Solarzellensystem Cu(In,Ga)(S,Se)₂ mit Hilfe von industrienahen Prozessen herzustellen. Der industrienahe zweistufige Herstellungsprozess beinhaltet das Aufbringen von metallischen Vorläuferschichten (Kathoden zerstäubung & thermische Verdampfung) und die Bildung des Chalkopyrit Absorbers in Folge des thermischen Ausheizschrittes. Der gezielte Einbau von Konzentrationsgradienten bestehend aus Ga/In und Se/S wird werkstoffseitig am Absorber strukturell analysiert und für die elektronischen Eigenschaften der Solarzelle optimiert. Zur Justage des Ga/In Profils werden sowohl das optimierte Temperaturprofil als auch ein Überangebot an Chalkogenen verwendet. Für die Einstellung des Se/S Profils im Chalkopyrit dienen das Ausgangsverhältnis sowie ein nachträglicher S Einbau (Erweiterung auf einen drei-stufigen Prozess). Um den Rückkontakt der Solarzelle vor den aggressiven Chalkogenen während des Prozesses zu schützen, wurde eine Mo-N Diffusionsbarriere entwickelt.

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Inscription

To my parents and to my brother.

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Nomenclature

The following abbreviations, symbols and codes are used in the thesis:

a:	Lattice constant	CGI:	Cu/(Ga+In) concentration ratio	
a(E):	Absorptance	CuGaS	Se: Copper Gallium	
Ag:	Silver	Silver		
Al:	Aluminium	CuGaS	Se ₂ : Copper Gallium	
α :	Absorption coefficient		Diselenide	
α_{eff} :	Effective absorption	χ :	Electron affinity	
- 5 5	coefficient	Cu(In,Ga)Se ₂ : Copper Indian		
AZO:	Al:ZnO		Gallium Diselenide	
	Back surface field	Cu(In,Ga)(S,Se) ₂ : Copper		
BSF:		•	Indium Gallium Disulfo	
β:	Linear \mathbf{E}_g grading degree		Selenide	
c:	Concentration	CuInS	2: Copper Indium Disulfide	
c_L :	Speed of light	CuInS	e ₂ : Copper Indium Diselenide	
CB:	Conduction band	Cu:	Copper	
CBD:	Chemical bath deposition	d_1 :	Back grading distance	
CE:	Co-evaporation	d_2 :	Front grading distance	

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D:	Diffusion coefficient of a diluted system	FWHM: Full width at half maximum		
d_c :	Critical distance	g:	Carrier generation rate	
DC:	Direct current	G:	Generation current	
Di:	Diode	Ga:	Gallium	
d_{hkl} :	Lattice spacing distance	GDOE	S: Glow discharge optical emission spectrometry	
D_n :	Carrier diffusion coefficient	CIXB	D: Crazing incidence X-ray	
DOS:	Density of states	GIMU	diffraction	
ξ:	Electric field	GGI:	Ga/(Ga+In) concentration	
ξ_{grad} :	Electric field in graded absorber		ratio	
		G_h :	Hole generation current	
e -:	Electron	G_n :	Electron generation	
EDX:	Energy dispersive X-ray		current	
	spectroscopy	\mathbf{h}^+ :	Hole	
E_F :	Fermi energy	\hat{H}_{eff}^{1e} :	Effective one electron Hamiltonian	
E_g :	Band gap	нтс.	High tomporature glass	
E_n :	One electron energy	пт G :	nigh temperature glass	
EMPA	: Swiss Federal	$h\nu$:	Photon energy	
	Laboratories for Materials	$\hbar\Omega$:	Phonon energy	
	Science and Technology	I:	Intensity	
EQE:	External quantum	IF:	Interface recombination	
η :	efficiency Efficiency	I_{mp} :	Current at maximum power point	
Fe:	Iron	In:	Indium	
FF:	Fill factor	IQE:	Internal quantum efficiency	

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I-V:	Current-voltage	J_{tot} :	Total current density
IXRD:	In-situ X-ray diffraction	K:	Pottasium
i-ZnO:	Intrinsic ZnO	k_B :	Boltzmann constant
J:	Current density	k_e :	Extinction coefficient
J_0 :	Reverse saturation current	<i>k</i> :	Scattering vector
0.01	lensity	L_{lpha} :	Absorption length
J_{bb} :	Photo current density,	L_n :	Electron diffusion length
	produced by black body	L_p :	Hole diffusion length
	radiation	M:	Diffusion coefficient
JCPDS	S: Joint Committee on Powder Diffraction	mc-Si:	Multicrystalline silicon
	Standards	MoN:	Molybdenum nitride
J_{dark} :	Dark current density	MoS_2 :	Molybdenum disulfide
J_{atom} :	Material flux	$MoSe_2$:	Molybdenum diselenide
J_{ph} :	er Photo current density,		2: Molybdenum disulfoselenide
	absorbed photons	\overline{n} :	Complex refractive index
J_{rec} :	Recombination current density	n_r :	Refractive index
		Na:	Sodium
J_{sc} :	Short-circuit current	N_A :	Acceptor density
	density	abla:	Gradient
J^{grad}_{sc} :	Short-circuit current density calculated by analytical approach	N_C :	DOS in the CB
		N_D :	Donor density
J_{sc}^{SCAPS} : Short-circuit current		n_{id} :	Diode ideality factor
	density calculated by SCAPS simulations	NREL:	National Renewable Energy Laboratory

N_V :	DOS in the VB	R_p :	Parallel resistance
n-ZnO:	n-doped ZnO	R_s :	Series resistance
Δn :	Excess charge carrier	RTP:	Rapid thermal processing
ODC:	density (electrons) Ordered defect compound	S/meta	d: S/(Cu+Ga+In) concentration ratio
PL:	Photoluminescence	S:	Sulfur
P_{mp} :	Maximum electrical power	SCAPS: Solar Cell Capacitance Simulator	
P_{opt} :	Power of incoming photons	SCR:	Space charge region
Ψ :	Electrostatic potential	Se/met	tal: Se/(Cu+Ga+In)
p-Si:	Polycrystalline silicon	,	concentration ratio
PV:	Photovoltaics	Se:	Selenium
ϕ_{sun} :	Photon flux from the sun	SEL:	Stacked elemental layer
$\phi(V,E)$:	Emitted photon flux,	SEM:	Scanning electron
	energy	SFG:	Standard float glass
Δp :	Excess charge carrier density (holes)	SIMS:	Secondary ion mass spectroscopy
Q:	Material quantity	SQ:	Shockley-Queisser
QNR:	Quasi neutral region	SRH:	Shockley-Read-Hall
R:	Recombination current	SSSe:	S/(S+Se) concentration
ho:	Charge distribution		ratio
R_h :	Hole recombination rate	<i>T</i> :	Temperature
R_n :	Electron recombination	Te:	Tellurium
	rate	T_S :	Sulfurization temperature

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μ_n :Electron mobility ϕ_n :One electron wave function μ_p :Hole mobility W_p :Watt peakVB:Valence band RRD :X-ray diffraction V_D :Built-in potential RRF :X-ray fluorescence v_e :Recombination velocity for electronsZn:Zinc V_{eff} :Effective potentialZSW:Centre for Solar Energy and Hydrogen Research Baden-Wuerttemberg v_{mp} :Voltage at maximum power pointZSW:Since for Solar Energy and Hydrogen Research Baden-Wuerttemberg	T_{Se} :	Selenization temperature	V_{oc} :	Open-circuit voltage
μ_p :Hole mobility W_p :Watt peakVB:Valence bandXRD:X-ray diffraction V_D :Built-in potential $Arref:$ X-ray fluorescence v_e :Recombination velocity for electronsZn:Zinc V_{eff} :Effective potentialZnS:Zinc blende v_{np} :Recombination velocity for holesZSW:Centre for Solar Energy ad Hydrogen Research Baden-Wuerttemberg	μ_n :	Electron mobility	ϕ_n :	One electron wave function
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v_h : Recombination velocity for ZSW : Centre for Solar Energy holes and Hydrogen Research V_{mp} : Voltage at maximum power point power point	V_{eff} :	Effective potential	ZnS:	Zinc blende
V_{mp} : Voltage at maximum power point Baden-Wuerttemberg	v_h :	Recombination velocity for holes	ZSW:	Centre for Solar Energy and Hydrogen Research
	V_{mp} :	Voltage at maximum power point		Baden-Wuerttemberg

Abstract

The aim of this work is to optimize the absorber for $Cu(In,Ga)(S,Se)_2$ thin film photovoltaic cells in order to get a higher efficiency of the energy conversion in the framework of a well-established industrial process. In general the industrial process belongs to the family of two-step processes which consist of sputtering and evaporating a stack of elemental layers and subsequent annealing in order to form the chalcopyrite structure of the absorber. The two-step process stays in competition with the more flexible co-evaporation process, which up to now has delivered the best laboratory cells reaching efficiencies up to 22.6%. The motivation for the two-step process may also be derived from the achievement of the best module conversion efficiencies of almost 18%. The Shockley-Queisser limit of single junction solar cells is around 33%. The difference of 11%to the best laboratory cells can be explained by optical reflection and internal recombination losses (Auger and Shockley-Read-Hall) and is mostly reflected in open-circuit voltage losses. In order to achieve higher opencircuit voltages, conduction band V-profiles are introduced in order to reduce space-charge region and interface (buffer) recombination as well as back contact recombination. Band gap grading is most easily achieved in the co-evaporation process. Due to the fact that the stacked elemental layer process is driven by interdiffusion of elements and compounds the precise control of these V-profiles is a challenge and alternative ways to control the band gap profile need to be found. To adapt the electronic band structure to the back electrode and the transparent window via a buffer layer, a certain gradient of Ga and S in the absorber is needed. In this work, optimized temperature profiles, optimal chalcogen amounts and a modified back contact have been identified as possible parameters for improvement of solar cell performance. In the field of band gap grading in Cu(In,Ga)Se₂ thin films a method was found to control the Ga/(Ga+In) profiles by the process temperature and the Se amount. This allows to optimize band gap profiles in the p-n junction solar cell with respect to more efficient charge carrier collection. With the help of numerical SCAPS simulations the optimized parameters of back and front surface grading properties can be calculated for the application in optimal process conditions. For the adjustment of S/(S+Se) profiles in $Cu(In,Ga)(S,Se)_2$ thin films, defined concentration gradients were established by applying chalcogens with a specific ratio before the annealing process. The conventional two-step process was furthermore modified by separating the annealing step into a pure selenization and a sulfurization phase in order to better adjust the S/(S+Se) profile. A large parameter field makes a detailed study of post sulfur diffusion processes possible. However, to adjust the chalcogen profile in the thin film it was found that the most important parameters are the sulfurization temperature and the sulfurization time in step two. Furthermore, a Mo-N layer was introduced as a novel barrier layer against chalcogen diffusion to the Mo back contact. A self-limited MoSe₂ layer growth on top of the Mo-N barrier layer with a defined thickness has been found, which does not hinder charge carrier collection at the back contact and additionally ensures a quasi-ohmic contact.